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5 **Solution composition and particle size effects on the dissolution and** 6 **solubility of a ThO₂ microstructural analogue for UO₂ matrix of** 7 **nuclear fuel**

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12 **Abstract**

13
14 The objective of this study was to investigate the dissolution rate of ThO₂ which was synthesised to approximate,
15 as closely as possible, the microstructure of UO₂ in a nuclear fuel matrix. The optimal sintering temperature for ThO₂
16 pellets was found to be 1750°C, which produced pellets with a microstructure similar to UO₂ nuclear fuel pellets,
17 with randomly oriented grains ranging in size from 10 to 30 μm. Dissolution was conducted using ThO₂ particles of
18 different size fractions (80 to 160 μm and 2 to 4 mm) in the presence and absence of carbonate in solutions with pH
19 from 2 to 8. Dissolution rates were calculated from Th released from the solid phase to solution. Particles of ThO₂
20 were also leached with 1 M HNO₃ at 80 °C in order to investigate the morphological changes at the particle surfaces.
21 The concentration of Th was found to be $\geq 10^{-9}$ mol/L at pH ≤ 4 , lower than the theoretical solubility of crystalline ThO₂.
22 At higher pH values, from 4 to 8, the measured concentrations (10^{-10} to 10^{-12} mol/L) were between the theoretical
23 solubility of ThO₂ and Th(OH)₄. Grain boundaries were shown to exert an influence on the dissolution of ThO₂
24 particles. Using high resolution aqueous solution analysis, these data presented here extend the current understanding
25 of Th solubility in solution.
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38 **1. Introduction**

39 Thorium dioxide (ThO₂) is isostructural to UO₂, sharing the same fluorite-type structure (space group Fm3m), making it
40 a useful structural analogue for spent nuclear fuel, which is predominantly composed of UO₂ (>95%). However, unlike
41 U^(IV)O₂, Th^(IV)O₂ is not redox active since Th has only one prevailing oxidation state, +4 [1]. The next generation
42 applications of nuclear energy have shown interest towards
43
44
45

46 thorium [2]. As a fuel, thorium has many beneficial proper-
47 ties, such as high fusion temperature, good sintering capa-
48 bility, resistance against radiation damage, greater abun-
49 dance in the Earth's crust compared to U, and the possibil-
50 ity for transmutation [2].

51 The preferred option worldwide for the long-term dis-
52 posal of spent nuclear fuel, and potentially future Th-based
53 nuclear fuels, is disposal in a deep geological disposal facil-
54 ity, several hundreds of metres below ground [3-4]. In this
55 environment, the release of Th and other radionuclides to
56 the geo- and bio-spheres will be controlled by the dissolu-
57 tion of the fuel by groundwater. Hence it is necessary to
58 understand the dissolution behaviour of the fuel and the
59 solubility of radionuclides in groundwater.

60
61 In the literature, the solubility values for ThO_2 , and also the
62 hydrolysis constants of thorium, show great discrepancies
63 e.g. Vandenberg et al. 2008, 2010 Neck and Kim, 2001 [5-
64 7]. The main reasons for these differences include: the
65 tendency of Th to undergo polynucleation and colloid for-
66 mation, its strong absorption to surfaces, and the low solu-
67 bility of Th^{4+} hydroxide and hydrous oxide. The presence of
68 complexing ligands like CO_3^{2-} [8] has also been shown to
69 increase the solubility of ThO_2 . These characteristics of
70 thorium together with the relatively low solubility of
71 $\text{ThO}_2(\text{cr})$ make solubility studies of ThO_2 challenging.

72
73 The solubility product values have been observed to vary
74 between ThO_2 (microcryst.) ($\log K^{\circ\text{sp}} = -53 \pm 0.5$) and
75 $\text{Th}(\text{OH})_4$ (am) ($\log K^{\circ\text{sp}} = -46.7 \pm 0.9$) [9-11] depending
76 on the crystallinity and crystallite size of the Th(IV) oxide
77 and hydroxide or oxyhydroxide phase. The predicted value,
78 according to the Equation of Schindler [12] and experi-
79 mental data, for ThO_2 (cr) is $\log K^{\circ\text{sp}} = -54.2 \pm 1.1$ [9]. In
80 addition to crystallinity, surface phenomena have been
81 mentioned as a possible factor affecting the solubility prop-
82 erties. Vandenberg et al. 2010 [6] combined investigation
83 of solid surfaces with leaching experiments, and isotopic
84 exchange methods to understand the discrepancy in solubil-
85 ity values and to describe the reversibility in the exchange
86 mechanism. They observed that the dissolution mainly
87 occurred at grain boundaries and showed variation between
88 different sites, indicating "local solubility" effects. In addi-
89 tion, the usage of ^{229}Th spiking revealed dynamic dissolu-
90 tion/precipitation reactions on the solid/solution interface.

91
92 Many solubility studies have been conducted with amor-
93 phous phases of ThO_2 [13-17]. The higher solubility of
94 amorphous-phase ThO_2 , compared to well-crystalline ThO_2 ,
95 simplifies the analysis of Th in liquid phase, because the
96 amorphous phases have higher solubility. Crystalline phas-
97 es of ThO_2 have also been studied [18-19]. Hubert et al.
98 [18] observed the effect of surface properties on the leachi-
99 bility of solid ThO_2 . Factors including specific surface area,
100 surface state and size of aggregates were found to have
101 influence on the apparent solubility. However, when the
102 leaching rate was normalized to surface area, it seemed to
103 be independent of the surface characteristics.

104

105 The aims of this study were to prepare crystalline ThO₂
106 pellets having a microstructure similar to that of spent UO₂
107 nuclear fuel pellets and to conduct dissolution experiments,
108 in order to further evaluate the relative solubility of ThO₂
109 phases. We report the initial release rate of Th during the
110 first 20 to 40 days of dissolution. Solubility studies were
111 extended to 100 to 120 days to gain a thorough understand-
112 ing of the solubility limit of different ThO₂ phases.

113 The first experiment series in the solubility and dissolution
114 rate studies was conducted with 2 to 4 mm particles in
115 0.1 M NaCl and 0.01 M NaCl (with 2 mM NaHCO₃) solu-
116 tions under atmospheric conditions. The second experiment
117 series was conducted with two particle sizes, 60 to 180 μm
118 and 2 to 4 mm, in 0.1 M NaCl and 0.01 M HNO₃ solutions
119 in an Ar glove box to exclude atmospheric carbon. In addi-
120 tion, several leaching experiments were run in 1 M HNO₃
121 solution and at 80 °C in order to observe surface changes
122 during dissolution in a relatively short time scale.

123 2. Experimental

124 2.1 Preparation of ThO₂ pellets

125 Thorium dioxide pellets were prepared to approximate the
126 microstructure of UO₂ fuel and CeO₂ analogues for UO₂
127 [19-21], with grain sizes in the range of 5 – 30 μm, with
128 random crystallographic orientation. The precursor material
129 was ThO₂ powder (BDH (British Drug House) Laboratory
130 Reagents Ltd., Lot No: G83757/541012), which was con-
131 firmed as pure ThO₂ by powder X-ray Diffraction using
132 STOE Cu-IP diffractometer, with a Cu Kα source (diffrac-
133 tion patterns were collected at $5 < 2\theta < 60^\circ$ at 2° min^{-1} ,
134 using a step size of 0.02°). 1g of ThO₂ powder was placed
135 within a 10 mm diameter hardened stainless steel die and
136 uniaxially pressed with a load of 100 MPa. The green den-
137 sity (i.e. compacted density) of the pressed compacts was
138 calculated prior to sintering by measuring the pellet mass
139 and geometry. Green ThO₂ pellets, placed on stabilised
140 zirconia setter plates, were sintered in triplicate for 4 hours
141 at temperatures between 1300 and 1750 °C in a standard air
142 atmosphere muffle furnace. Pellets were heated and cooled
143 with a ramp rate of 5 °C min⁻¹ and held at the sintering
144 temperature for 4 hours.

145
146 The sintered density of the pellets was measured using
147 geometric and water immersion (Archimedes) methods. All
148 density measurements were performed in triplicate.

149
150 For surface analysis only (i.e. not dissolution experiments)
151 ThO₂ pellet samples were polished to a 1 μm finish using
152 SiC paper and diamond paste. A final mechanico-chemical
153 etch was performed using a 0.06 μm colloidal silica solu-
154 tion. In order to develop a grain boundary texture at the
155 surface of the pellets, annealing was performed at 90 % of
156 the sintering temperature, at a ramp rate of 5 °C min⁻¹ and
157 held at the annealing temperature for 1 hour. Pellets were
158 imaged using an optical microscope. Analysis of the crys-

159 tallographic orientation and grain size distribution of pellets
160 was performed using Electron Backscatter Diffraction
161 Analysis (EBSD) (Oxford Instruments) in conjunction with
162 a Sirion Field Emission SEM. EBSD maps of 100 μm^2 were
163 obtained at an accelerating voltage of 20 kV and a 0.5 μm
164 step size. Grain orientation analysis was performed on >
165 10000 grains, using HK Channel 5 software (Oxford In-
166 struments).

167 **2.2 Fragmentation of the pellets**

168 For the experiments, the ThO_2 pellets were crushed using a
169 percussion mortar. Particles between 2 and 4 mm were
170 selected with tweezers and washed four times in isopropanol
171 followed by gravitational settling in an attempt to re-
172 move adhering fine fragments. This was not successful, so
173 the particles were soaked one at a time on isopropanol fol-
174 lowed by ethanol and then dried in a desiccators.

175
176 The intact pellets of ThO_2 (~ 20 g) were fragmented to
177 smaller particle sizes. In order to avoid contamination by
178 the milling media, the electrodynamic fragmentation
179 method [22-23] was selected to comminute the pellets into
180 the required particle sizes. The principles of the method
181 can be briefly described, as follows: Electrical energy in
182 the form of repeated high voltage pulses is applied to the
183 samples immersed in a dielectric process liquid. Dielectric
184 liquids, like water, have a high dielectric strength, when
185 the voltage rise time is kept below 500 ns. As such, the
186 discharges are forced to occur through the immersed ma-
187 terial. Plasma channels and explosions were generated
188 inside the pellets and the resulting shockwaves produced
189 fracturing and physical breakdown.

190
191 Fragmentation was conducted using a batch scale selfFrag
192 instrument available at the Research Laboratory of the
193 Geological Survey of Finland.

194
195 The commercialized 1.7 tons instrument designed for the
196 c.a. 1 kg sample was used. Approximately 20 grams of
197 ThO_2 pellets were subjected to a two stage treatment. The
198 process liquid used was regular tap water. After the first
199 fragmentation the sample was classified using 0.2 mm net
200 sieve, and the remaining oversized fraction (ca. 14 g) was
201 re-fragmented. The process parameters used for both the
202 first and second fragmentation procedure were: number of
203 pulses 400, e-gap 10mm, freq. 3 and voltage 120-140 kV.

204
205 The fragmented material was washed using tap water and
206 any iron contamination from the electrode was removed
207 using hand magnets. The final product for use in dissolution
208 experiments was sieved to a particle size fraction of 80 to
209 160 μm .

210 **2.3 Dissolution experiments**

211 The first experiment series of solubility and dissolution rate
212 studies were conducted with 2 to 4 mm particles in 0.1 M
213 NaCl and 0.01 M NaCl (with 2 mM NaHCO_3) solutions
214 under atmospheric conditions (solutions were prepared

215 from suprapure (99.99%, Merck) NaCl and NaHCO₃
216 (ACS, Reag. Ph Eur, Merck KGaA) in MilliQ-water). Exper-
217 iments were conducted in triplicate at room temperature
218 (23 ± 1 °C) for up to 115 days. Approximately 300 mg of
219 crushed ThO₂ particles were placed in a 60 ml high density
220 polypropylene vessel with 50 ml of leaching solution. The
221 second experiment series was conducted with two particle
222 sizes (60 to 180 µm and 2 to 4 mm), both in 0.1 M NaCl
223 and 0.01 M HNO₃ (prepared from concentrated HNO₃,
224 ULTREX II by J.T Baker) within an Ar glove box at
225 25 ± 1 °C. The duration of these experiments, conducted in
226 duplicate, was 93 days. Approximately 150 mg of particles
227 were placed in a 60 ml vessel with 50 ml of leaching solu-
228 tion. Polypropylene vessels were used for experiments with
229 0.01M HNO₃, and high density perfluoroalkoxy Teflon
230 vessels were used for 0.01M NaCl solutions, in order to
231 decrease the potential sorption of Th to the reaction vessel
232 under near neutral conditions.

233
234 In the first experiment series, conducted under atmospheric
235 conditions, sampling was performed at 0, 1, 3, 6, 15, 24, 31
236 41, 48, 79, 100 and 115 days by withdrawing a 2.5 ml sam-
237 ple. Each sample was ultrafiltered with a Pall Mall filtration
238 device with 10 kD (~1 nm) molecular cut off, using centrifi-
239 gation (6000 rpm, 1 h). In addition, some samples were
240 taken and not filtered. The pH was measured directly from
241 the test solution with ROSS combination electrode at the
242 beginning of the experiment and after 35 days of reaction
243 time.

244
245 Sampling of the experiment series under an Ar atmosphere
246 was conducted at 0, 1, 2, 5, 7, 9, 14, 21, 30, 43, 57, 70 and
247 89 days. Both non-filtered and ultrafiltered samples (as
248 above) of 2.5 ml were taken. The pH was measured in solu-
249 tions under Ar glovebox conditions using a ROSS combina-
250 tion electrode, directly from the reaction vessel within eve-
251 ry second sampling.

252
253 The concentration of ²³²Th in non-filtered and ultrafiltered
254 samples was analysed with a High Resolution sector Field
255 Inductively Coupled Plasma-Mass Spectrometer (HR-ICP-
256 MS, Element2 by ThermoScientific). Standard solutions
257 with known concentrations were diluted from AccuTrace™
258 Reference Standard SQS-01. A control sample for analysis
259 was prepared from standard CLMS-1 solution by SPEX.
260 All the blank, standard and control samples contained a
261 known amount of an internal standard, indium. Analyses of
262 ²³²Th were performed with HR-ICP-MS in low resolution
263 mode (R ≈ 300). The detection limit for thorium was calcu-
264 lated as six times the standard deviation of the ion counts
265 obtained for the sample blanks (ten replicates), divided by
266 the sensitivity of the 1 µg/L standard solution of Th. The
267 detection limit for thorium was therefore found to be be-
268 tween 1·10⁻¹² mol/L and 4·10⁻¹² mol/L depending on the
269 solution matrix and daily efficiency of the instrument. The
270 uncertainty of the Th analyses increased up to 10 % when
271 the measured concentration was close to the detection limit.
272 With higher concentration the uncertainty of the analysis
273 was a few percent.

274

275 The dissolution rate ($r_{232\text{Th}}$ [$\text{mol g}^{-1}\text{s}^{-1}$]) for ThO_2 was
 276 calculated by using equation (1)
 277

$$278 \quad r_{232\text{Th}} = \frac{V}{m} \frac{dc_{232\text{Th}}}{dt} \quad (1)$$

279 where V is the volume of the solution [L], m is the mass [g]
 280 of ThO_2 and dc/dt is the slope (linear fit) determined from
 281 the evolution of ^{232}Th release as function of time [$\text{mol L}^{-1}\text{s}^{-1}$].
 282 This analysis is similar to that applied to consideration of
 283 mineral dissolution elsewhere e.g. in Rozalen et al, 2008,
 284 Malmström M., 1996, Brady and Walther 1990 [24-26].
 285

286 The surface area was calculated by assuming that the ThO_2
 287 particles are cubes (which is not true as is commonly
 288 known see e.g. [27]) and using the measured density of
 289 ThO_2 approximately 9 g/cm^3 (~93 % of the theoretical 9.86
 290 g/cm^3) [28].
 291

$$\rho_{\text{ThO}_2} = \frac{m_{\text{ThO}_2}}{V_{\text{tot,ThO}_2}}$$

$$V_{\text{grain}} = d^3$$

$$N_{\text{grain,ThO}_2} = \frac{V_{\text{tot,ThO}_2}}{V_{\text{grain}}}$$

$$A_{\text{grain,ThO}_2} = 6d_{\text{grain,ThO}_2}^2$$

$$SA_{\text{tot,ThO}_2} = N_{\text{grain,ThO}_2} A_{\text{grain,ThO}_2} = 6 \frac{m_{\text{ThO}_2}}{\rho_{\text{ThO}_2}} \frac{1}{d_{\text{ThO}_2}} \quad (2)$$

297 **2.4 Characterization of leached ThO_2 surfaces** 298 **with SEM**

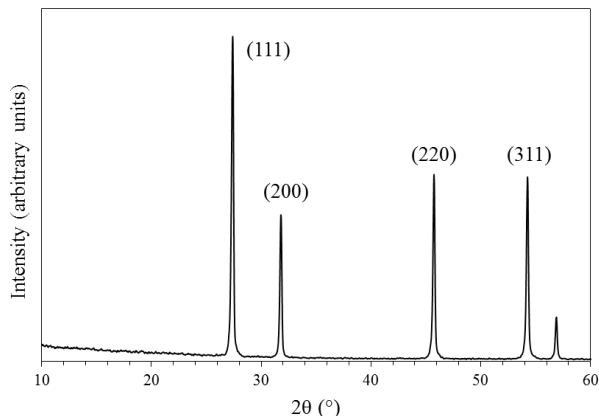
299 The ThO_2 particles were subjected to leaching experiments
 300 in acidic conditions and at elevated temperatures in order to
 301 monitor the evolution of surface morphology during dissolution.
 302 For detailed characterisation of the pellet morphology during
 303 dissolution, see Corkhill et al. [29]. Surface characterisation
 304 was performed on particles in the 80 to 160 μm size fraction,
 305 following particles were used for surface characterization after
 306 immersion in a 1 M HNO_3 (~ pH 1) solution at 80 °C for 2 and
 307 4 weeks under Ar atmosphere. Prior to heating the reaction
 308 vessels were placed in a sealed steel container under Ar atmosphere
 309 of the glove box to maintain argon atmosphere. Surfaces of
 310 ThO_2 subject to this treatment were studied with SEM (JEOL
 311 JSM-900LV with Oxford Instruments) using an accelerating voltage
 312 of 20 kV and beam size of 10 μm .
 313

314 **3. Results and Discussion**

315 **3.1 The properties of sintered ThO_2 pellets**

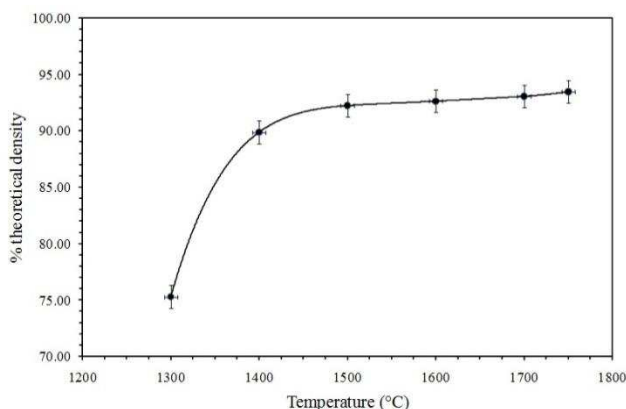
316 Figure 1 shows the X-ray diffraction analysis of ThO_2 powder,
 317 confirming the purity of the starting material. The characteristics
 318 for a suitable UO_2 fuel analogue include a grain

319 size on the order of 8–15 μm , a sintered density of
320 > 95.5 % theoretical density, and randomly orientated
321 grains. The sintering temperature was varied in order to
322 determine the optimal sintering conditions to create such
323 characteristics. Figure 2 shows the densities of the sintered
324 pellets as a function of sintering temperature.



325
326 **Fig. 1.** XRD pattern for thorium dioxide powder used to synthe-
327 sise UO_2 -fuel analogue pellets for dissolution experiments.

328
329 The density of the pellets increased with increasing sinter-
330 ing temperature up to 1750 $^{\circ}\text{C}$ (Fig. 2); this temperature
331 gave a density of 93 % of the theoretical density, which is
332 slightly below optimal for a UO_2 analogue. Heating at
333 higher temperatures caused a loss of pellet integrity, there-
334 fore 1750 $^{\circ}\text{C}$ was the highest sintering temperature evaluat-
335 ed. The increase in density with increasing sintering tem-
336 perature is consistent with decreasing porosity and increas-
337 ing grain size, as shown by optical microscope images in
338 Figure 3. The grain size was found to increase from 2 – 10
339 μm at 1650 $^{\circ}\text{C}$ to 10 – 30 μm at 1750 $^{\circ}\text{C}$ (Table 1). Sintering
340 at 1750 $^{\circ}\text{C}$ was, therefore, found to give the optimum grain
341 size and density achievable. EBSD analysis was performed
342 to determine grain size and crystallographic orientation.
343 The average grain size was found to be 13 μm (based upon
344 analysis of >10000 grains). Figure 4 shows that synthesis at
345 1750 $^{\circ}\text{C}$ produced grains that were randomly orientated in
346 the {111}, {100} and {101} crystal planes
347 .



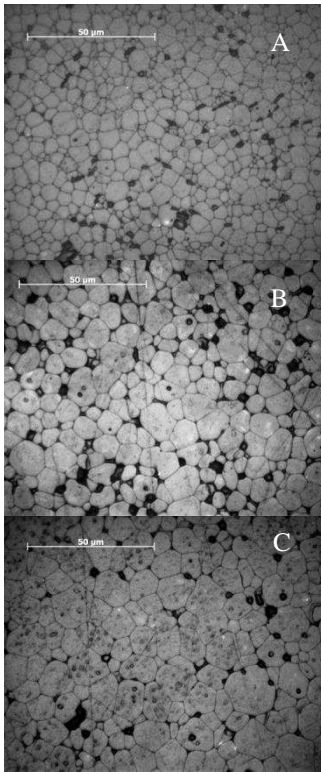
348
349 **Fig. 2.** Density of the sintered ThO_2 as function of sintering
350 temperature.

351

352

353

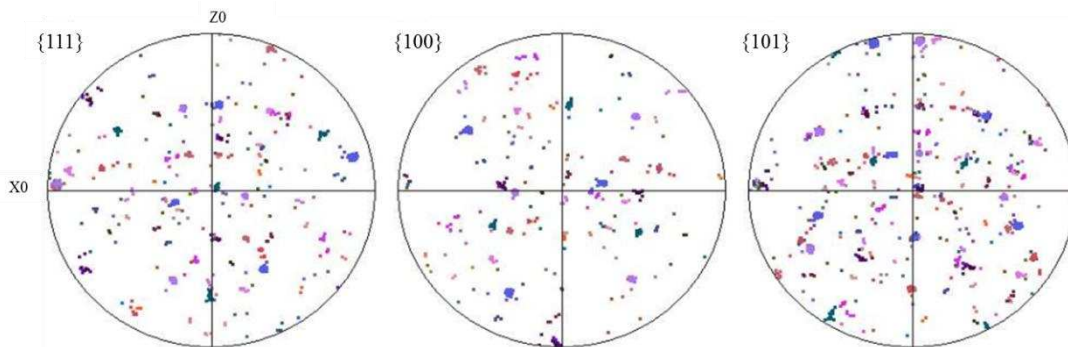
354 **Fig. 3.** Optical microscopy images of the ThO₂ pellet surface
 355 sintered at 1600 (A), 1700 (B), and 1750 °C (C).



356 **Table 1.** The grain size of ThO₂ pellets as a function of sintering
 357 temperature.

Sintering temperature (°C)	Grain size (μm)
1600	2-10
1700	5-20
1750	10-30

358



359

360 **Fig. 4.** Representative crystal orientations of a pressed pellet of
 361 ThO₂ sintered at 1750 °C and annealed at 1500 °C to develop
 362 grain structure. Pole figures show the random orientation of the
 363 grains in the {111}, {100} and {101} crystal planes.

364

365 **3.2 Particle characteristics after selfFrag HV** 366 **pulsing**

367 Fragmentation of the ThO₂ pellets produced randomly
 368 shaped ThO₂ particles, as confirmed by optical and electron
 369 microscopy (Figs. 5 and 6). Similar randomly broken grains

370 typically result from applying conventional comminution
371 techniques e.g. grinding. However, adhered fines, normally
372 seen in SEM images showing products from conventional
373 comminution, were not observed, due to the pulse fragmen-
374 tation procedure. Adhere-particle free surfaces can be re-
375 garded as ideal samples for the dissolution experiments
376 undertaken in this study.

377 The particles exhibited two main textures: those with a
378 grain boundary texture and those without, as shown in Fig-
379 ure 6. Grain boundaries were formed in these particles
380 through two processes. Firstly, prior to fragmentation, sur-
381 faces of the original pellets developed a grain boundary
382 texture through high temperature annealing. It may also be
383 possible that such surfaces originate from the pulse frag-
384 mentation process, which forces liquid through grain
385 boundaries to break the particles apart, leaving behind sev-
386 eral surfaces with a grain boundary texture.

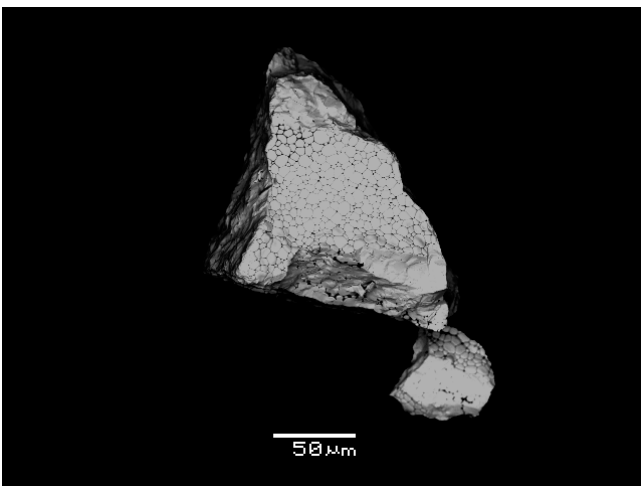
387



388

389 **Fig. 5.** An image of 80 to 160 μm ThO_2 particles taken with a
390 biocular stereomicroscope.

391



392

393 **Fig. 6.** SEM image of 80 to 160 μm ThO_2 particle crushed using
394 the Selfag pulse fragmentation technique (330 x magnification).

396 3.3 Dissolution experiments

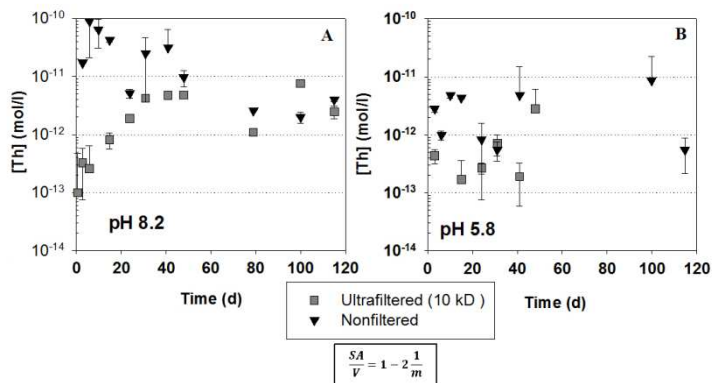
397 3.3.1 ThO₂ dissolution under atmospheric conditions

398

399 Figure 7 shows the dissolution data for ThO₂ particles (2 - 4
400 mm) leached in air in 0.1 M NaCl or in 0.01M NaCl with 2
401 mM NaHCO₃. In the solution containing carbonate
402 (Fig. 7a), a maximum concentration of Th of $\sim 10^{-10}$ mol/L
403 was measured in non-filtered samples after 6 days of disso-
404 lution. After 40 days the concentration plateaued at approx-
405 imately 10^{-12} mol/L. The ultrafiltered samples reached a
406 concentration of between 10^{-11} and 10^{-12} mol/L at 40 days,
407 preceded by a steady increase in concentration from 10^{-13}
408 mol/L. The initial pH value of 8.4 decreased slightly to pH
409 8.2 during the first 35 days of the experiment. In the 0.1 M
410 NaCl solution, shown in Figure 7b, the increase of the
411 measured concentration was not obvious; the concentra-
412 tions in both non-filtered and ultrafiltered samples are scat-
413 tered (Fig 7b) most probably due the concentrations occur-
414 ring close to the analytical detection limit ($1 \cdot 10^{-13}$ mol/L).
415 The concentration of Th in the ultrafiltered samples de-
416 creased below this detection limit of after 48 days sam-
417 pling. The initial pH of the 0.1 M NaCl increased from 5.4
418 to pH 5.8 during the first 35 days of the experiment.

419

420 When comparing the results of the two experiments, the
421 solubility and the dissolution rates were greater in the solu-
422 tion containing carbonate, ($1 \cdot 10^{-13}$ mol dm⁻³ d⁻¹), likely due
423 to, the formation of Th-containing carbonate/hydroxide
424 complexes. In the absence of carbonate (0.1 M NaCl solu-
425 tion) there was no clear trend, however, if the dissolution
426 rate is estimated from ultrafiltered samples, the obtained
427 value was found to be $2 \cdot 10^{-14}$ mol dm⁻³ d⁻¹, an order of
428 magnitude less than that in the carbonate-containing solu-
429 tion. In the oxic conditions utilised in these experiments
430 (which were conducted in air), dissolved carbon dioxide
431 from the atmosphere may have had a slight solubility in-
432 creasing effect in the bi-carbonate-free solution. The in-
433 creasing effect of carbonate complexation on the solubility
434 of thorium in each of the applied experimental conditions
435 was previously evaluated by geochemical modelling
436 (PHREEQC) [30]. A second set of experiments was there-
437 fore conducted under inert Ar atmosphere, to exclude such
438 effects.



439
440

441 **Fig. 7.** Evolution of Th concentrations in ultrafiltered and non-
442 filtered experiments of 2 – 4mm ThO₂ particle dissolution in (A)
443 0.01 M NaCl with 2 mM NaHCO₃ and (B) in 0.1 M NaCl, both
444 under atmospheric conditions, at a temperature of 23°C.

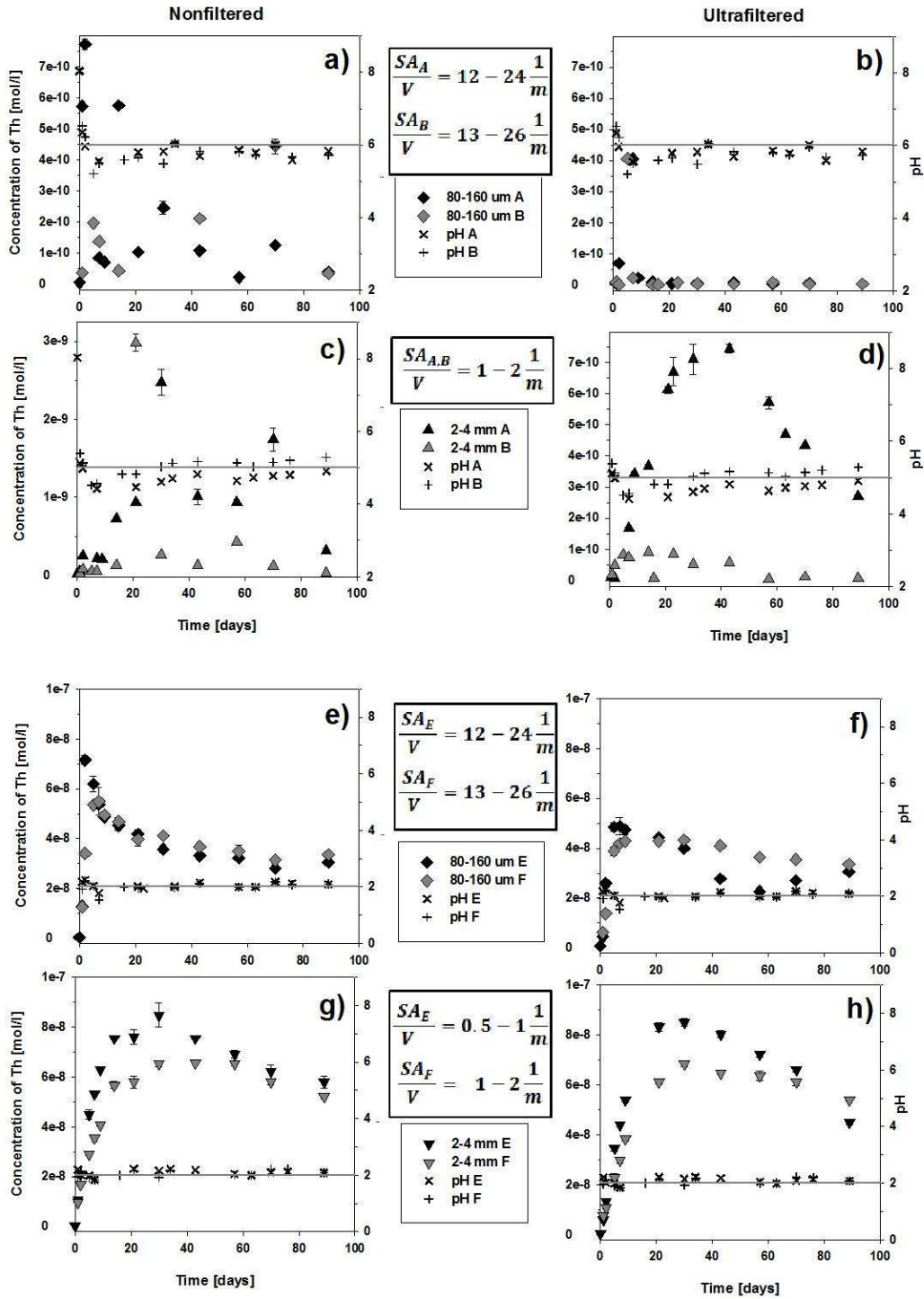
445

446 3.3.2 ThO₂ dissolution in anaerobic conditions

447

448 The dissolution data resulting from the leaching of 80 to
449 160 μm and 2 – 4 mm particles of ThO₂ in 0.1 M NaCl
450 under an Ar atmosphere are shown in Figure 8. Unfiltered
451 samples arising from the smaller size fraction (80 to 160
452 μm) demonstrated a relatively rapid initial increase in Th
453 concentration to $\sim 8 \cdot 10^{-10}$ mol/L after 2 days of dissolution,
454 followed by a progressive decrease to $4 \cdot 10^{-11}$ mol/L by 89
455 days (Fig 8a). Under the same conditions, ultrafiltered sam-
456 ples (Fig. 8b) gave a maximum Th concentration of $\sim 4 \cdot 10^{-10}$
457 mol/L after 7 days of leaching. After 9 days of dissolu-
458 tion, the concentration of Th decreased below $1 \cdot 10^{-11}$ mol/L
459 and remained at $\sim 2 \cdot 10^{-12}$ mol/L after 57 days.

460



461
 462 **Fig. 8.** The evolution of Th concentration and pH in 0.1 M NaCl
 463 (a, b, c, d) and in 0.01 M HNO₃ (e, f, g, h) with particle sizes 80 to
 464 160 μm (a, b, e, f) and 2 to 4 mm (c, d, g, h) at 25 °C. The results
 465 of the Non-filtered samples are given on the left and the ULTRA-
 466 filtered ones on the right. The solid lines were sketched to guide
 467 the eye to follow the pH values. Filtrations, solutions and particle
 468 sizes presented here are also shown with references to this Fig. in
 469 Table 2. (symbols A and B (E and F) in legend boxes refer to the
 470 parallel experiments).
 471

472 The maximum concentrations of Th were reached after 23
 473 days of dissolution ($3 \cdot 10^{-9}$ mol/L and $7.4 \cdot 10^{-10}$ mol/L for

474 non-filtered and ultrafiltered samples, respectively). This
475 suggests that dissolution and precipitation / sorption kinet-
476 ics are also slower in these samples, which is likely related
477 to the lower surface area resulting from the larger particle
478 size; the saturation limit is reached more slowly because
479 there is less surface area available for sorption and precipi-
480 tation of secondary Th compounds.

481 The initial pH of the 2- 4 mm ThO₂ dissolution experiments
482 was found to decrease rapidly from pH 8 to between pH 5
483 and pH 6, where it remained for the duration of the experi-
484 ment. This unexpected result is likely due to the leaching of
485 H⁺ from incompletely rinsed reaction vessels.

486 Parallel, ultrafiltered experiments conducted for the 2 – 4
487 mm particle size (designated A and B, Fig. 8d) showed
488 considerable variation in concentration, by almost one order
489 of magnitude. This may be due to the influence of pH in the
490 test solutions; experiment A had a pH value of 4.8, while
491 experiment B gave a pH value of 5.3. At the conclusion of
492 the experiment, the Th concentrations were $7 \cdot 10^{-12}$ mol/L
493 (experiment A, pH 4.8) and $3 \cdot 10^{-10}$ mol/L (experiment B,
494 pH 5.3). While the difference in pH was only small, these
495 observations confirm the fact that the solubility of ThO₂ is
496 highly dependent on the H⁺ activity in the solution [9-11].

497

498 3.3.3 ThO₂ dissolution as a function of particle size

499

500 To investigate further the influence of particle size on the
501 dissolution of ThO₂, experiments were conducted in 0.01 M
502 HNO₃, where dissolved Th concentrations far exceeded the
503 analytical detection limit of Th, allowing for an accurate
504 determination of the effects of particle size on dissolution
505 rate.. For 80 to 160 μm particles of ThO₂ (Figs. 8e and f),
506 the concentration in the non-filtered samples increased
507 rapidly to $5 \cdot 10^{-8}$ mol/L after 5 days of dissolution. Similar-
508 ly, in the ultrafiltered samples, the Th concentration in-
509 creased to $4 \cdot 10^{-8}$ mol/L over the same time period. After
510 this initial rapid dissolution, the concentration of Th de-
511 creased to levels between $3 \cdot 10^{-8}$ and $4 \cdot 10^{-8}$ mol/L for the
512 duration of the 90 days experiment, in both non-filtered and
513 ultrafiltered samples.

514

515 For the 2 to 4 mm ThO₂ particles, the initial dissolution rate
516 was slower than that of the smaller particle size. The maxi-
517 mum concentration of Th was reached after 30 days of
518 dissolution, giving values of $6.5 \cdot 10^{-8}$ mol/L and $8.5 \cdot 10^{-8}$
519 mol/L for duplicate samples (Fig. 8g -h). The final concen-
520 tration of Th was $\sim 5 \times 10^{-8}$ mol/L after 90 days, which was
521 slightly higher than the final concentration arising from the
522 smaller particle size ($\sim 3 \times 10^{-8}$ mol/L) (Figs. 8e-f). This
523 suggests that particles with a greater surface area (e.g. 80 –
524 160 μm size fraction) reach lower maximum concentrations
525 than particles with lower surface area (e.g. 2 – 4 mm parti-
526 cles) a result of more surface sites available for sorption of
527 secondary precipitates. Such precipitates attenuate further
528 dissolution of the surface. A similar effect was observed for
529 particles subject to dissolution in 0.1 M NaCl (described
530 above). However, unlike the dissolution experiments con-
531 ducted in 0.1M NaCl, where dissolution rates were clearly
532 affected by filtration and hence, the presence of colloidal

533 species, the evolution of Th concentration in 0.01M HNO₃
534 was independent of the sampling method; the concentra-
535 tions of the ultrafiltered samples were very similar to the
536 non-filtered ones. This indicates that secondary phases or
537 colloids did not play a significant role in these experiments.
538 One possible explanation may be effects arising from high
539 energy sites on the surface, for example grain boundaries
540 (shown in Fig. 6). The larger particles comprise surfaces
541 with more grains than the smaller particles, therefore they
542 have a greater number of grain boundaries. In a recent study
543 [29] it was found that grain boundaries of ThO₂ and CeO₂
544 analogues for UO₂ fuel dissolved very rapidly in 0.01M
545 HNO₃. Therefore, the higher maximum concentrations
546 found for the large particles may be due to dissolution of
547 grain boundaries

548

549 **3.3.4 Dissolution rates of ThO₂**

550 The initial dissolution rates of ThO₂ under the experimental
551 conditions investigated were calculated from the evolution
552 of Th concentration during the period of release of Th into
553 solution, which took between 5 to 30 days, depending on
554 the liquid phase in question.

555 Calculated rates were obtained from linear regression of
556 these data, and are given in Table 2. The results clearly
557 show the pH dependence of the dissolution rate and the
558 increasing effects of carbonate concentration and particle
559 size. The dissolution rates were found to be greatest at low
560 pH and in the presence of carbonate. Small differences
561 were also observed between parallel experiments and in the
562 comparison between results calculated from non-filtered
563 and ultrafiltered samplings.

564

565 The presence of carbonates is known to increase the solu-
566 bility of ThO₂, which can have a significant effect on the
567 behaviour of Th(IV) in natural groundwaters. One order of
568 magnitude increase in [CO₃²⁻] has been shown to increase
569 the solubility of hydrous ThO₂(am) by up to five orders of
570 magnitude [31]. Two mononuclear carbonate complexes of
571 Th(IV), Th(OH)₃CO₃⁻ and Th(CO₃)₅⁶⁻, have been reported
572 [31, 32]. Considerably higher Th(IV) solubility in the
573 Th(IV)-H₂O-CO₃²⁻ system, indicated the presence of highly
574 charged pentacarbonate species [31]. Using thermodynamic
575 modelling, Kim et al. [8] reported ternary complexes such
576 as Th(OH)₃CO₃⁻, Th(OH)₂(CO₃)₃⁴⁻, Th(OH)(CO₃)₅⁵⁻,
577 Th(OH)₂CO₃(aq), Th(OH)₂(CO₃)₂²⁻, Th(OH)₄(CO₃)₂²⁻, which
578 are the most probable predominant aqueous Th(IV) species
579 under many natural conditions [8].

580

581 At high pH close to pH 11.2, the higher concentration of
582 OH⁻ ions has been found to enhance the formation of
583 Th(OH)₄, and thus decrease the proportion of carbonate
584 complexes in solution. Therefore, the measured solubility
585 of Th(IV) in alkaline solutions is very close to the solubility
586 in carbonate-free solutions [8].

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Table 2. The initial dissolution rates [$\text{mol/g}^{-1}\text{s}^{-1}$] calculated for samples A and E and their parallel samples B and F, according to Equation (1).

Atmosphere and filtration	Solution	pH	Particle size	Test A (E) slope, dc/dt [$\text{molL}^{-1}\text{day}^{-1}$]	Test A (E) rate [$\text{mol g}^{-1}\text{s}^{-1}$]	Test B (F) slope, dc/dt [$\text{molL}^{-1}\text{day}^{-1}$]	Test B (F) rate [$\text{mol g}^{-1}\text{s}^{-1}$]	Data in Figure
Air (filtered)	0.01M NaCl (2mM NaHCO ₃)	8	2-4mm	1.00E-13	2.00E-19			Fig 7B
Air (filtered)	0.1M NaCl	6	2-4mm	2.00E-14	4.10E-20			Fig 7A
Air	0.01M NaCl (2mM NaHCO ₃)	8	2-4mm	5.00E-12	1.10E-17			Fig 7B
Air	0.1M NaCl	6	2-4mm	3.00E-13	6.10E-19			Fig 7A
Argon (filtered)	0.1M NaCl	6	80-160um	5.40E-11	2.10E-16	2.90E-11	1.10E-16	Fig 8b
Argon (filtered)	0.1M NaCl	5	2-4mm	2.90E-11	1.00E-16	8.40E-12	2.10E-17	Fig 8d
Argon (filtered)	0.01M HNO ₃	2	80-160um	6.70E-09	2.70E-14	5.80E-09	2.20E-14	Fig 8f
Argon (filtered)	0.01M HNO ₃	2	2-4mm	6.30E-09	2.30E-14	4.40E-09	1.70E-14	Fig 8h
Argon	0.1M NaCl	5	80-160um	4.20E-10	1.70E-15	1.50E-10	5.60E-16	Fig 8a
Argon	0.1M NaCl	4	2-4mm	4.40E-11	1.50E-16	9.20E-12	2.30E-17	Fig 8c
Argon	0.01M HNO ₃	2	80-160um	3.10E-08	1.30E-13	9.30E-09	3.50E-14	Fig 8e
Argon	0.01M HNO ₃	2	2-4mm	6.50E-09	2.40E-14	4.40E-09	1.70E-14	Fig 8g

593

Atmosphere and filtration	Solution	pH	Particle size	Test A (E) slope, dc/dt [$\text{molL}^{-1}\text{day}^{-1}$]	Test A (E) rate [$\text{mol g}^{-1}\text{s}^{-1}$]	Test B (F) slope, dc/dt [$\text{molL}^{-1}\text{day}^{-1}$]	Test B (F) rate [$\text{mol g}^{-1}\text{s}^{-1}$]
Air (filtered)	0.01M NaCl (2mM NaHCO ₃)	8	2-4mm	1.0E-13	2.0E-19		
Air (filtered)	0.1M NaCl	6	2-4mm	2.0E-14	4.1E-20		
Air	0.01M NaCl (2mM NaHCO ₃)	8	2-4mm	5.0E-12	1.1E-17		
Air	0.1M NaCl	6	2-4mm	3.0E-13	6.1E-19		
Argon (filtered)	0.1M NaCl	6	80-160 μm	5.4E-11	2.1E-16	2.9E-11	1.1E-16
Argon (filtered)	0.1M NaCl	5	2-4mm	2.9E-11	1.0E-16	8.4E-12	2.1E-17
Argon (filtered)	0.01M HNO ₃	2	80-160 μm	6.7E-09	2.7E-14	5.8E-09	2.2E-14
Argon (filtered)	0.01M HNO ₃	2	2-4mm	6.3E-09	2.3E-14	4.4E-09	1.7E-14
Argon	0.1M NaCl	5	80-160 μm	4.2E-10	1.7E-15	1.5E-10	5.6E-16
Argon	0.1M NaCl	4	2-4mm	4.4E-11	1.5E-16	9.2E-12	2.3E-17
Argon	0.01M HNO ₃	2	80-160 μm	3.1E-08	1.3E-13	9.3E-09	3.5E-14
Argon	0.01M HNO ₃	2	2-4mm	6.5E-09	2.4E-14	4.4E-09	1.7E-14

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If reader is interested in calculating the dissolution rate to units [$\text{mol m}^{-2}\text{s}^{-2}$], one can e.g divide the values by $3.3\text{cm}^2/\text{g}$ in cases of 2mm particle size and $80\text{cm}^2/\text{g}$ in cases of $80\mu\text{m}$.

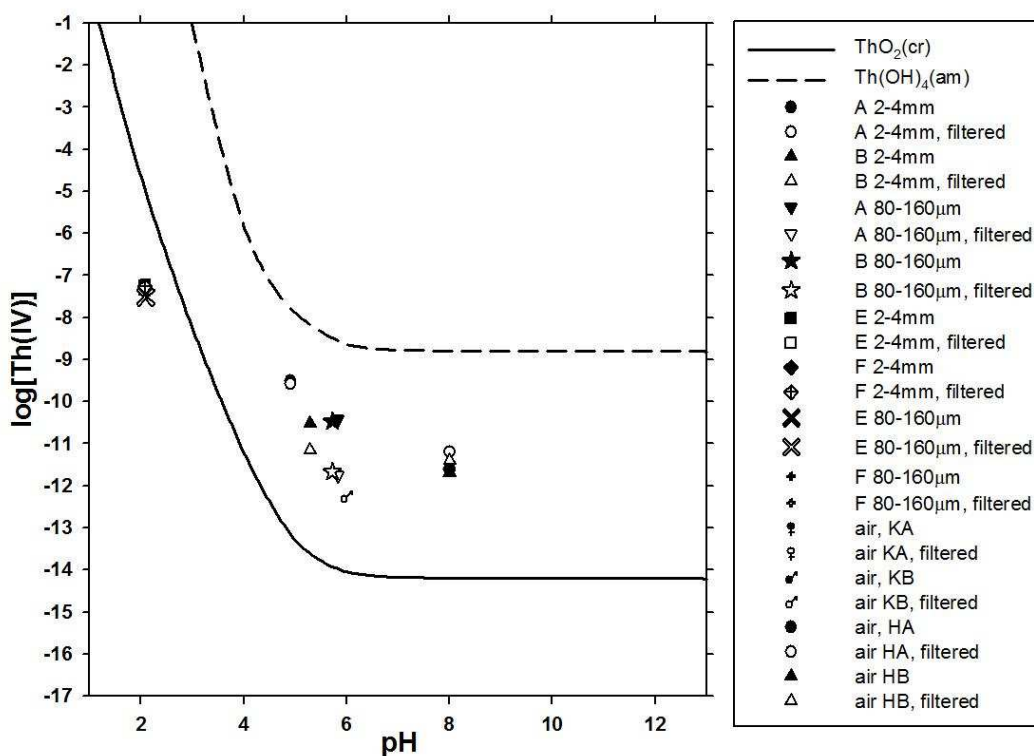
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3.3.4 Solubility of ThO₂

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The solubility levels at the end of the experiments were compared with the equilibrium data of ThO₂ and Th(OH)₄, which were taken from the Thermochemie database [NEA/TDB] (Fig. 9). It should be noted that, for example, in the experiment with 2 to 4 mm particles in 0.1 M NaCl under Ar, there was still some decrease in Th concentration at the end of the 90 days experiment. Thus, the solubility level had probably not been stabilized to a steady state during the experiments. The solubilities measured at pH 2 were at a lower level than would be expected from the thermodynamics. Similar observations have been made by

610 Neck et al. [11]. In their undersaturation dissolution exper-
 611 iments, conducted also with crystalline ThO_2 in acidic con-
 612 ditions, the solubilities stayed below the theoretical solu-
 613 bility of ThO_2 . In their experiments they also showed that
 614 the value of the solubility product was dependent on the
 615 crystallite size. In this study, the solubility values at a pH
 616 range of 4 to 8 can be plotted between the theoretical solu-
 617 bilities of crystalline ThO_2 and amorphous Th(OH)_4 (Fig.
 618 9). Compared to the studies discussed by Neck et al. 2003
 619 and Vandenborre et al. 2010 [11, 6], the values presented
 620 here are at an even lower level. This could be an effect of
 621 the crystallinity and relatively high sintering temperature of
 622 the ThO_2 utilized in these studies, however the usage of a
 623 sector field ICP-MS enabled an extremely accurate analyses
 624 of dissolved Th concentrations (down to a concentration of
 625 10^{-12} mol/L), which is lower than achievable with conven-
 626 tional quadrupole ICP-MS MS, used in most other studies
 627 of Th solubility.
 628



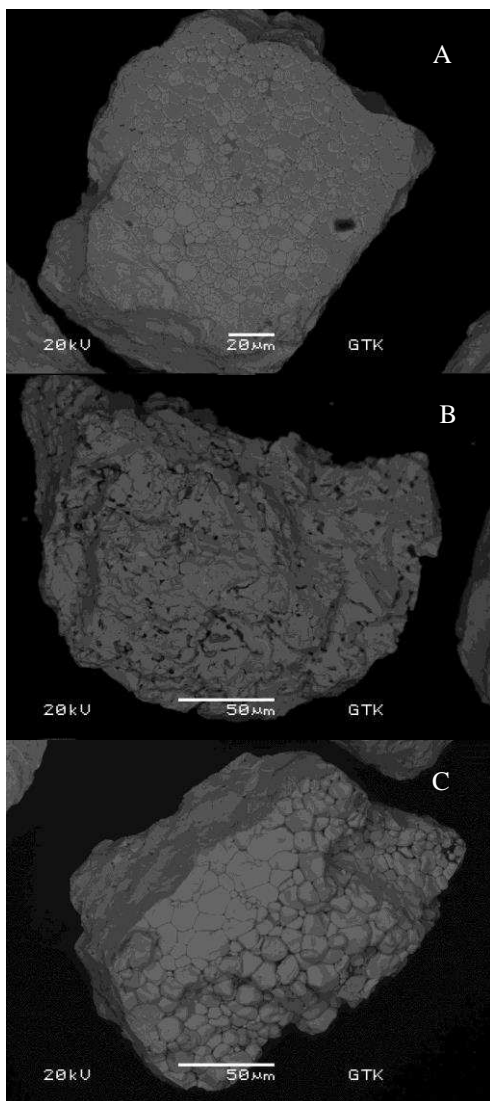
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 630
 631 **Fig. 9.** The solubility levels of Th after ~100 days of dissolution in
 632 a suite of media investigated in the current work, compared to the
 633 Thermochemie equilibrium data for ThO_2 and Th(OH)_4
 634 [NEA/TDB].

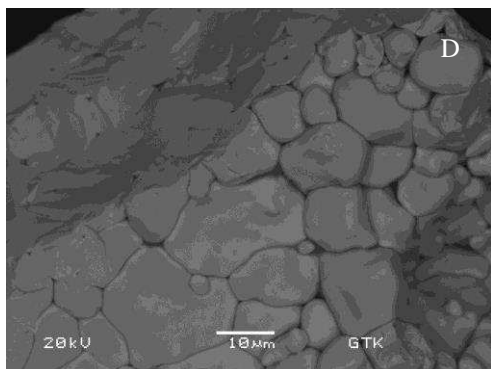
635 3.4 SEM imaging of leached ThO_2 surfaces

636 The SEM images of the HNO_3 leached ThO_2 particles re-
 637 vealed that the nature of the dissolution/precipitation phe-
 638 nomenon varied among the particles present in same solu-
 639 tion. Some particles did not appear different from the fresh
 640 and unreacted particles (see Figures 10a and Figure 6, re-
 641 spectively). However, in a single particle, some surfaces

642 showed textures indicative of grain boundary dissolution
643 (Figs. 10b – d), while others appeared to host a “blanket” of
644 secondary phases precipitated from solution. In the former,
645 grain rounding and widening of the grain boundaries was
646 observed (Fig. 10b).

647
648 These results are in agreement with Corkhill et al. [29] who
649 found that during dissolution in 0.01M HNO₃, grain bound-
650 aries of ThO₂ and CeO₂ analogues for UO₂ fuel dissolved
651 rapidly, giving rise to enhanced initial dissolution rates.
652 Furthermore, they found that surfaces with a grain bounda-
653 ry texture (like those shown in Fig. 10a) dissolved at much
654 less rapid rate than surfaces with no grain boundaries, such
655 as those that would be found on the surface of a sintered
656 pellet. This may be the cause of the different particle mor-
657 phologies described in the current investigation.
658





659 **Fig. 10.** BSE images of 80 to 160 μm ThO₂ particle after 4 weeks
 660 of leaching with 1 M HNO₃ at 80°C. A) a leached particle resem-
 661 bling a fresh one, B) a particle showing deeper dissolution, which
 662 has deformed the grain boundaries and formed some cavities into
 663 the ThO₂ particle. C and D are taken from this particle, which
 664 showed both areas of dissolution and no reaction.

665 4. Conclusions

666 ThO₂ pellets were synthesised to resemble the
 667 microstructure of the UO₂ matrix of nuclear fuel pellets.
 668 The pellets sintered at 1750 ° C had randomly oriented
 669 crystals and a grain size ranging from 10 to 30 μm.

670 Dissolution experiments were performed to study the effect
 671 of carbonate complexation on these ThO₂ samples. It was
 672 observed that the solubility and dissolution of ThO₂
 673 increased in the presence of carbonate. The solubilities and
 674 dissolution rates of ThO₂ were also dependent on the H⁺
 675 activity of the solution. The acidic conditions were found to
 676 increase the solubility compared to near-neutral and basic
 677 conditions, in which sparingly soluble hydrolysed species
 678 of Th were the dissolution-controlling factor. In these
 679 experiments the concentration of Th ($\geq 10^{-9}$ mol/L) at pH
 680 ≤ 4 was slightly lower than the theoretical solubility of
 681 crystalline ThO₂. HR-ICP-MS, with magnetic sector field
 682 capability, allowed the analyses of dissolved Th
 683 concentrations of sparingly soluble ThO₂ phases down to
 684 10^{-12} mol/L. At higher pH values, from pH 4 to pH 8, the
 685 measured concentrations (10^{-10} to 10^{-12} mol/L) were
 686 between the theoretical solubility of ThO₂ and Th(OH)₄.

687 Particle size was also found to have an effect on the
 688 solubility. The smaller particles with greater surface area
 689 seem to have enhancing effect on initial dissolution, and
 690 sorption/precipitation reactions. In agreement with recent
 691 results [29], it is hypothesized that high energy surface sites
 692 may play a role in the relatively rapid initial release of Th
 693 observed here. The results of the surface analyses indicated
 694 that the initial surface condition may also have some effect
 695 on the dissolution processes, with grain boundaries likely
 696 playing an important role.

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 702 tation, of the ThO₂ pellets and SEM imaging of the leached parti-

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